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# Impact of  $H_2O$  on  $CO_2$  adsorption and co-adsorption: Mechanism and high-performance adsorbents for efficient  $H_2O$ -CO<sub>2</sub> capture



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#### A R T I C L E I N F O

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#### ABSTRACT

Physical adsorbents used for  $CO_2$  capture from air directly and wet  $CO_2$  flue gas have limited effectiveness due to competitive adsorption of H<sub>2</sub>O and CO<sub>2</sub>. Therefore, it is crucial to understand the impact of H<sub>2</sub>O on CO<sub>2</sub> adsorption and investigate the co-adsorption mechanism in order to enhance the screening, design, and optimization of carbon capture systems. Herein, heat treatments are performed on commercial NaX zeolites to modify the composition of the O-T-O (T = Al, Si) framework. Our systematic investigation based on thermogravimetric analysis/differential scanning calorimetry (TG-DSC), infrared spectroscopy (IR), solid-state nuclear magnetic resonance (SSNMR), and density-functional theory (DFT) calculation, the connection between the structure of NaX zeolite and H2O and CO2 adsorption/desorption is established. The heat-induced changes in the O-T-O (T = Al, Si) framework produce movements of the loaded  $Na<sup>+</sup>$  ions. Hydrogen bonds formed by agglomerated adsorbed  $H_2O$  constitute the primary source of desorption heat, which adversely affects  $CO_2$  mass transfer and adsorption. Our findings illustrate that modifying the distribution of Na<sup>+</sup> ions in the O-T-O (T = Al, Si) framework can reduce the CO<sub>2</sub> adsorption energy and prevent the formation of water agglomerates. Our findings provide insights into the design of efficient and high-capacity adsorbents for CO<sub>2</sub> capture that can be operated in the presence of  $H_2O$ .

#### **1. Introduction**

Since the concentration of  $CO<sub>2</sub>$  in the atmosphere is lower than that in industrial flue gases, traditional chemical carbon capture methods using alcohol amines, ammonia, and ionic solutions to capture  $CO<sub>2</sub>$  from ambient air may not be applied directly to flue gases. The potential environmental impact associated with the lifecycle of these methods including equipment, labor, operation, and maintenance costs often outweighs the benefits of capturing  $CO<sub>2</sub>$  from the atmosphere [\[1,2\]](#page-10-0). Despite recent advances in the reduction of the overall cost and energy requirements, these methods may not provide tangible benefits for enterprises and society [\[3\].](#page-10-0) Therefore, it is crucial to explore low-cost, high-efficiency direct air carbon capture technologies. However, the development of affordable, effective, energy-efficient, environmentally friendly, and safe  $CO<sub>2</sub>$  adsorbents for large-scale implementation is a daunting task [\[4\]](#page-10-0).

The direct air carbon capture (DAC) process not only captures  $CO<sub>2</sub>$ selectively, but also adsorbs a significant amount of  $H_2O$  from air. Given the larger concentration of  $H_2O$  compared to  $CO_2$  in ambient air, the coadsorption of  $H_2O$  and  $CO_2$  cannot be overlooked [\[5\].](#page-10-0) The presence of  $H<sub>2</sub>O$  can either enhance or hinder the  $CO<sub>2</sub>$  capture capacity depending on the type of sorbents, and may even impact the long-term stability [\[6\]](#page-10-0). The influence of  $H<sub>2</sub>O$  co-adsorption on amine-functionalized sorbents has been studied in recent years [\[7,8\]](#page-10-0). However, physical adsorption without amine groups offers a more straightforward approach than complex chemical adsorption processes and allows better control of the

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adsorption selectivity. Madden et al.  $[9]$  have evaluated the CO<sub>2</sub> capture capacity of ten physisorbents and found that H2O impedes adsorption, which can be mitigated by precise control of the pore size and pore chemistry.

In sorbent-based direct air capture (DAC) systems, in order to minimize the impact of  $H_2O$  on  $CO_2$  capture, one approach is to incorporate an additional desiccant bed prior to the  $CO<sub>2</sub>$  adsorption bed for moisture removal [\[10\]](#page-10-0). However, this method presents certain challenges. Firstly, inclusion of extra desiccant beds increases the system complexity and cost and additional equipment and piping are required to facilitate desiccant circulation and regeneration. This not only escalates the equipment and operational expenses, but also increases the potential risk of system malfunctioning. Secondly, the desiccant bed requires additional space, potentially limiting the feasibility and adaptability of the system in practical applications. Furthermore, when selecting a desiccant and considering its regeneration process, other factors such as cost, sustainability, and environmental impact must be taken into account [\[11\]](#page-10-0).

Co-capture reduces the amount of required adsorbent and device size, resulting in a smaller overall system volume and weight. This integration also facilitates easier regeneration of the adsorbent, enables multiple cycles, and improves the sustainability and economy of the adsorption process. The recovery of  $H_2O$ , an important resource in  $CO_2$ utilization, daily life, and industrial applications, holds great significance and efficiency in the joint retrieval of  $CO<sub>2</sub>$  [\[12\].](#page-10-0) In the future, as DAC systems become more widespread, recovering  $H<sub>2</sub>O$  from air can address operational challenges, particularly in regions with water scarcity such as deserts and arid coastal areas [\[13\]](#page-10-0). Additionally, advancements in photocatalysis, photothermal catalysis, and other technologies offer the opportunities to convert  $H_2O$  and  $CO_2$  into highquality fuels [14–[16\]](#page-10-0). Therefore, future DAC systems require adsorbents that can achieve equilibrium adsorption while efficiently separating  $H<sub>2</sub>O$  and  $CO<sub>2</sub>$ .

Historically, DAC systems have predominantly utilized organic amine-loaded chemical adsorbents and alkaline solutions for  $CO<sub>2</sub>$  capture  $[17]$ . While these methods can effectively separate  $CO<sub>2</sub>$  from other impurities during adsorption, the reverse reaction during desorption and subsequent adsorbent/solution regeneration necessitate higher energy inputs (operating temperature *>*373.15 K). Porous materials utilized for physical adsorption of CO<sub>2</sub> rely on low-energy van der Waals forces and electrostatic forces [\[18\]](#page-10-0). In terms of environmental friendliness (aminefree) and low-energy/low-cost operation, physical adsorbents are highly desirable [\[19,20\]](#page-10-0). Among them, zeolites have gained considerable attention in gas separation due to their abundant porous structure, cost effectiveness, and well-established production processes [\[21\]](#page-10-0). Compared with other solid physical adsorbents (4A zeolite [\[22\]](#page-10-0), ZSM-5 zeolite  $[23]$ , metal-based MOFs  $[24]$ , physical membrane  $[25]$ , etc.), the large porous domain of NaX zeolite limits its adsorption selectivity for  $CO<sub>2</sub>$  and H<sub>2</sub>O. While the adsorption selectivity of NaX zeolite for  $CO<sub>2</sub>$ and  $H<sub>2</sub>O$  is limited due to its large pore size, its exceptional adsorption capacity and successful commercial implementation should not be overlooked. NaX zeolite delivers exceptional performance in purification systems for air separation while capturing more than 99.99 % of  $CO<sub>2</sub>$  and H<sub>2</sub>O from air [\[26,27\]](#page-10-0).

NaX zeolite, which is commercially mature and widely used in industry, is used as an adsorbent. On the one hand, it has high capacity, high adsorption selectivity, high hydrothermal stability and regeneration. On the other hand, zeolites are widely present minerals on the earth, and the preparation of NaX zeolite is relatively easy. This reduces production costs and increases sustainability. Compared with other adsorbents used for  $CO<sub>2</sub>$  capture, the structure of zeolite can be controlled through a variety of methods, thereby changing its pore size and surface properties, making it more suitable for the carbon capture process under specific conditions. For direct air carbon capture with low CO2 concentration, NaX zeolite is undoubtedly one of the most economical and practical adsorbents.

Water is frequently found in atmospheres containing carbon dioxide gas, including air and industrial emission sources such as flue gas, biogas and natural gas. As far as adsorption is concerned, a wide variety of carbon dioxide separation materials exhibit many different behaviors in the presence of water. Depending on the material, water may promote or hinder  $CO<sub>2</sub>$  adsorption, increase or decrease the rate of  $CO<sub>2</sub>$  absorption, or even affect the structural integrity of the adsorbent. Studying the competitive behavior of  $CO<sub>2</sub>$  and  $H<sub>2</sub>O$  on adsorbent materials can help design more effective adsorbents and improve the efficiency of carbon capture.

Current research activities on zeolite capture of  $H_2O$  and  $CO_2$  can be divided into two approaches. The first approach focuses on adjusting the hydrophobicity of zeolites specifically for  $CO<sub>2</sub>$  capture [\[28\].](#page-10-0) However, the hydrophobic layer on zeolites affects the kinetics rather than the thermodynamics of adsorption. Consequently, the equilibrium capacity for water adsorption does not change, but the time required to reach equilibrium is significantly longer. The second approach explores the water adsorption properties of zeolites for applications such as air drying and water capture in certain regions  $[29]$ . The inclusion of  $Al^{3+}$  into the zeolite framework creates polar sites in the zeolite structure to render them hydrophilic [\[30\]](#page-10-0). Therefore, the hydrophilic/hydrophobic properties of zeolites can be optimized by adjusting the Si/Al ratio in the structures and large Si/Al ratios result in stronger hydrophobic characteristics of zeolites. Furthermore, optimizing the hydrophilic/hydrophobic properties of zeolites can enhance gas adsorption [\[31\]](#page-10-0). Coadsorption of  $CO<sub>2</sub>$  and  $H<sub>2</sub>O$  is influenced by the interactions between  $CO<sub>2</sub>$  and  $H<sub>2</sub>O$  binding sites and their respective characteristics. The effects of  $H_2O$  on  $CO_2$  adsorption thus depend on these factors. A better understanding of the impact of  $H_2O$  on  $CO_2$  adsorption can assist researchers in analyzing the influence moisture when designing and selecting adsorbent materials. Moreover, by better understanding the co-adsorption mechanisms, researchers can optimize the chemical and physical properties of materials to enhance the adsorption capacity and selectivity of  $CO<sub>2</sub>$  and moisture.

Herein, the influence of  $H_2O$  on  $CO_2$  adsorption and co-adsorption mechanism of commercially available NaX zeolites are investigated ([Fig. 1](#page-2-0), Table S1 and Fig. S1-2).  $CO<sub>2</sub>$  and H<sub>2</sub>O exhibit similar adsorption behavior including shared or similar adsorption sites. During coadsorption of H2O and CO2, the faster mass transfer of H2O leads to preferential occupation of the co-adsorption sites. Additionally, agglomeration of  $H_2O$  molecules impedes diffusion and adsorption of  $CO<sub>2</sub>$ . A systematic investigation on the temperature-induced structural changes of zeolites and their impact on adsorption of  $H_2O$  and  $CO_2$  is performed. As  $Na<sup>+</sup>$  ions are commonly found at adsorption sites, their migration in the zeolite framework and subsequent redistribution influence the adsorption characteristics for  $H_2O$ . Our results also reveal that alteration of the zeolite framework affects the adsorption energy of  $CO<sub>2</sub>$  revealing a close relationship between structural changes and  $CO<sub>2</sub>$ adsorption.

#### **2. Methodology**

#### *2.1. Material and characterization*

The commercial NaX zeolites were purchased from Jianlong Micro-Nano New Materials Co., Ltd., Luoyang, China (Table S1). In the heat treatment, the commercial NaX zeolite was sintered in a muffle furnace at different temperatures (433.15, 523.15, 573.15, 623.15, 673.15, 723.15, 773.15 and 823.15 K) for 4 h. X-ray diffraction was performed on the Bruker D8 Advance with Cu K<sub>α</sub> radiation ( $\lambda = 1.540598$  Å), a step size of 2◦/min, and a range of 5–90◦. The structure was examined by field-emission scanning electron microscopy (ZEISS), and the specific surface, pore volume, pore diameter, and free energy were determined using a fully automatic specific surface and porosity analyzer (Autosorb-IQ-MP, Quanta, USA) in a high-purity nitrogen environment at 77 K. The sample was degassed at a temperature of 393.15 K in vacuum for 12 h

<span id="page-2-0"></span>

**Fig. 1.** NaX zeolite structure with different scales: (a) Macro appearance and see supporting information for details; (b–c) Surface morphology of the sample prepared by grinding the spherical particles in NaX zeolite which is octahedral with a rough surface and the binder used in molding making the surface uneven; (d–f) 3D NaX crystal structure at different angles.

before acquiring the adsorption isotherms and adsorption/desorption curves for  $CO<sub>2</sub>$  under high-purity carbon dioxide at 298 K.

The PQ001 nuclear magnetic resonance analyzer produced by Shanghai Numei Electronic Technology Co., Ltd. was used to determine the 2T characteristics of the transverse relaxation time. The magnetic strength, magnet temperature, and coil diameter were 0.55T, 32 ◦C, and 25 mm, respectively. There was a glass cylinder inside the coil and the sample was spread on the bottom of the glass cylinder. The transverse relaxation time 2T was measured by the CPMG pulse sequence. The NMR 2T relaxation spectra were obtained after inversion of the acquired signal by the inversion software MultiExp Inv Analysis that came with the instrument.

The infrared spectra were acquired from the zeolite on the Perkin Elmer Spectrum3 Fourier transform infrared spectrometer (FTIR). Potassium bromide compression was adopted and the scanning range was 4000 to 400  $\text{cm}^{-1}$ . The XPS spectra were obtained on the Thermo Fisher Scientific ESCALAB 250Xi using a monochromatic Al  $K_{\alpha}$  source with an energy of 1487.20 eV. The  $^1\mathrm{H}$  solid-state nuclear magnetic spectra were acquired on the Bruker Advance III HD at 400 MHz and the sample was first adsorbed and saturated in the  $N_2$  containing water vapor.

#### *2.2. Analysis of gas adsorption/desorption*

The temperature-programmed adsorption/desorption behavior was analyzed by thermogravimetric analysis on the Setline Instrument (Fig. S3). The temperature of the sample was programmed from room temperature (293.15 K) to 453.15 K at a ramping rate of 10 K/min under flowing nitrogen (40 mL/min) and then cooled to 313.15 K after holding the temperature for 60 min to achieve complete desorption from the sample. Different gas flows were applied and the adsorption capacity for  $CO<sub>2</sub>$  was in 400 ppm  $CO<sub>2</sub>/N<sub>2</sub>$  at a flow rate of 40 mL/min. The adsorption capacity for  $CO_2 + H_2O$  (feedback in mass form) was determined in 400 ppm  $CO<sub>2</sub>/N<sub>2</sub>$  + water vapor (controlling the water bath temperature to obtain a relative humidity of 40 % RH) at a flow rate of 40 mL/min. The desorption behavior including the concentration change of  $CO<sub>2</sub>$  and  $H<sub>2</sub>O$ was analyzed by a flue gas analyzer (Shenzhen Watersell Co., Ltd.). The desorption exotherms were analyzed by differential scanning calorimetry on the Setline instrument. Different heating rates were used to study the form (physical or chemical bond) of the adsorbate in the adsorbent and the order of desorption.

#### *2.3. Density-functional theory calculations*

The unit cell unit of NaX was constructed based on the information in the literature. The lattice constant of 1 cubic NaX was 25.03 Å and the unit cell of NaX consisted of 556 O, 104 Si, 88 Al, and 41 Na atoms. NaX is described in the supporting information. Based on these initial structures, Castep's geometric optimization was performed to derive the most stable configuration of NaX and then the optimized structures were used in the simulations. During optimization, the atom positions and shape of the unit cells are not fixed. Since the unit cell of each NaX phase was large, only the smallest repeating unit was optimized and the Gamma point was sampled. The unique localized functional CA-PZ and local density approximation (LDA) methods were used to correct the unit cell structure. The membrane potential was Ultrasoft (ultrasoft), and the energy cutoff (cutoff energy) was 450 eV [\[32\].](#page-10-0)

Molecular dynamics simulations were carried out using the Forcite program in Materials Studio 8.0, and the force field parameters were taken from the literature. NPT (combined isothermal-isobaric) simulations were performed at 0.0 GPa and 50, 100, 200, 300, 433.15, 523.15, 573.15, 623.15, 673.15, 723.15, 773.15, and 823.15 K. The temperature and pressure were controlled by the Andersen thermostat and Parrinello barometer, respectively. The simulation time was 500 ps and the data collected in the last 200 ps were used to analyze the thermal expansion behavior.

The structural changes after the heat treatment were probed using Quench in the Forcite program and all parameters were the same as aforementioned. The number of iterations was 500, and the energy, force and displacement convergence were set to  $1 \times 10^{-4}$  Ha,  $5 \times 10^{-3}$ Ha and  $5 \times 10^{-5}$  Å, respectively. Among the obtained structural models at different temperatures, a model that was similar to the molecular dynamics simulation structure at the corresponding temperature was selected [\[33\].](#page-10-0)

The adsorption behavior was simulated using the Sorption program in Materials Studio 8.0. The structural models obtained by heat treatment at different temperatures were subjected to the (111) cleaved surface. A vacuum layer of 10 Å was subsequently established. The adsorption isotherm and fixed pressure task were used on the cut surface, respectively, and  $CO<sub>2</sub>$  and  $H<sub>2</sub>O$  passing the geometric optimization in Castep were adsorbed by the Metropolis method. The starting pressure was 0.001 kPa and the ending pressure was 0.01 and 0.1 kPa, <span id="page-3-0"></span>respectively.

#### **3. Results and discussion**

#### *3.1. Adsorption mechanism of CO2 and H2O of NaX*

By analyzing the Differential Scanning Calorimetry (DSC) spectrum, information about the thermal properties of the sample can be obtained, such as melting point, crystallization temperature, reaction heat, glass transition, etc. [\[34,35\]](#page-11-0). DSC is widely used in materials science, drug research and development, food industry and other fields. DSC is performed to determine the physical properties of  $CO<sub>2</sub>$  and  $H<sub>2</sub>O$  adsorption on NaX. Depending on the adsorption type (physical or chemical adsorption) of different adsorbates in the desorption process, the heat of desorption varies [\[36\]](#page-11-0). The heat peaks reflect the respective adsorption types of  $CO<sub>2</sub>$  and  $H<sub>2</sub>O$  of NaX.



**Fig. 2.** Desorption of H2O on NaX for different heating rates: (a) 0.5 ℃/min; (b) 1 ℃/min; (c) 1.5 ℃/min; (d) 2 ℃/min; (e) 5 ℃/min; (f) 10 ℃/min.

#### *3.2. Adsorption of H2O of NaX*

First, NaX zeolite is selected for saturation adsorption by  $N_2$  containing water vapor. using TG-DSC apparatus (TG mass stays unaltered). The desorption of  $H_2O$  is an endothermic process since the adsorbent's process of adsorbing  $H_2O$  is exothermic. As a result, using DSC technology at various heating rates, the endothermic peaks of  $H<sub>2</sub>O$  adsorbed by NaX during the desorption process may be examined, as shown in [Fig. 2a](#page-3-0)–f. Endothermic peaks usually rise, while exothermic peaks fall. The width of the thermal peak can also provide information about the reaction rate. The NaX desorption heat flow curves of  $H_2O$  at heating rates of 0.5 ℃/min, 1 ℃/min, 1.5 ℃/min, 2 ℃/min, 5 ℃/min and 10 ℃/min were obtained respectively. It can be found that as the heating rate increases, the time for NaX to desorb H2O decreases. When the heating rate is higher, the instantaneous high thermal energy can perform multiple types of H2O desorption simultaneously. When the heating rate is lower than 2 ◦C/min, the peak shape and peak valley of the heat flow peak can be observed more clearly. According to the fitting results, two obvious heat flow peaks appear when H2O adsorbed in NaX is desorbed.

By integrating the area under the thermal peak, the total amount of heat absorbed during the desorption process can be obtained. [Fig. 2](#page-3-0)  shows the desorption heat value of the same mass of NaX after saturated H2O adsorption, with an average value of 855.70 J/g. Since thetotal heat of desorption is basically the same, the change in the heating rate can change the time of desorption heat for each type of adsorbed materials. Analysis of the peak time of the desorption peak is obvious. According to the control of the desorption heating rate, the adsorbent of two different adsorption forms can be separated. The separation principle is equivalent to the working principle of gas phase and liquid chromatography. The adsorption type is determined by the force of each adsorption form (physical or chemical adsorption, adsorption strength, etc.). The starting time and ending time of heat absorption are obtained by fitting the peaks of different types of adsorbed substances. The separation of desorption is realized by using the time difference between the start and end.

Due to the different distribution positions of adsorbed  $H_2O$  in the adsorbent, different NaX adsorption sites have different adsorption energies for H2O. Exploring the adsorption form of H2O in NaX will help us analyze the distribution of  $H<sub>2</sub>O$  adsorption sites and the adsorption energy of H2O at different sites. Low Field-Nuclear Magnetic Resonance (LF-NMR) is mainly used in the study of physical properties, using relaxation time and diffusion coefficient parameters to reflect the dynamic information of molecules in the sample. The strength and size of the 1H signal in water is an important basis for effectively determining the distribution and binding form of water. To analyze the form of  $H_2O$ in NaX, the transverse relaxation time is determined by low-frequency NMR. The magnitude of the transverse relaxation time represents the strength of the binding energy of water in the zeolite, and the smaller the transverse relaxation time, the stronger the binding energy with water [\[37\]](#page-11-0). Different peaks represent different bound forms of moisture. The peaks with relaxation times between 1 and 10 ms indicate the presence of weakly chemically bound water and those in the range of 10–100 ms indicate physically bound water. The pristine sample shows weakly chemically bound water (2.8–3.4 ms), indicating that the weakly chemically bound Na<sup>+</sup> formed with water is by adsorption ( $Fig. S4$ ). This means that it has a relatively strong weak chemical adsorption energy compared to other samples, and desorption consumes more energy.

#### *3.3. Co-adsorption of CO2 and H2O of NaX*

In order to further analyze the adsorption/desorption mechanism of CO2 and H2O in NaX when CO2 and H2O are co-captured. After determining the adsorption form of H2O in NaX for preferential diffusion mass transfer, the desorption heat flow after NaX adsorbed  $CO<sub>2</sub>$  and  $CO<sub>2</sub>$ -H2O respectively after saturation was also obtained based on DSC technology. There are 3 heat flow peaks and 6 heat flow peaks in the

desorption process of  $CO<sub>2</sub>$  and  $CO<sub>2</sub>-H<sub>2</sub>O$  on NaX, respectively [\(Fig. 3a](#page-5-0),b). During co-adsorption of  $CO_2$  and  $H_2O$  on NaX, free  $H_2O$  in the form of hydrogen bonds accounts for the largest composition (peak 4). The weak physical adsorption of  $CO<sub>2</sub>$  with H<sub>2</sub>O and O-Si-O is the weakest force in NaX (peak 1 and peak 2). Therefore,  $CO<sub>2</sub>$  first absorbs heat and desorbs in peak 1 and with the appearance of peak 2, some of the adsorbed  $H_2O$ desorbs (peak 3). Free  $H<sub>2</sub>O$  (peak 4) continues to absorb heat for desorption. With the continuous supply of heat, the weak chemical adsorption of  $H_2O$  with Na<sup>+</sup>, CO<sub>2</sub> and O-Al-O begins to break (peaks 5 and 6). As shown in [Fig. 3](#page-5-0)c–e, based on the intermolecular force and adsorption capacity, the analysis of  $CO<sub>2</sub>$  and  $H<sub>2</sub>O$  in NaX is described in the following (the peak number shows the order in which the force breaks):

- (1) When NaX only adsorbs  $H_2O$ , Na<sup>+</sup> forms weak chemical adsorption with OH– in H<sub>2</sub>O, and H<sup>+</sup> in H<sub>2</sub>O also forms weak chemical adsorption with the acidic skeleton O-Al-O. Adsorption also occurs due to the hydrogen bonds between  $H_2O$  and  $H_2O$  (binding energy − 5.72 kcal/mol, Table S2). Therefore, as adsorption proceeds, there will be a stream of physically adsorbed water in the form of "agglomeration" in NaX. When NaX only adsorbs  $CO<sub>2</sub>$ , since Na<sup>+</sup> and O-T-O (T = Al, Si) form a balanced electric field, CO2 adsorbs near them by van der Waals force and electrostatic force forming weak physical adsorption. Since there is no active adsorption between  $CO<sub>2</sub>$  and  $CO<sub>2</sub>$  (binding energy 0.14 kcal/ mol), in the same skeleton, the adsorption capacity of  $H<sub>2</sub>O$  is smaller than that of  $CO<sub>2</sub>$ .
- (2) When  $H_2O$  and  $CO_2$  are adsorbed by NaX at the same time in a humid air environment, the concentration of  $CO<sub>2</sub>$  (400 ppm) is lower than that of water vapor in air (RH 40 %). At a low partial pressure, H2O has lower adsorption energy in NaX, and the mass transfer rate of  $H_2O$  is higher than that of  $CO_2$  (Figs. S5a–d). Therefore,  $H_2O$  reaches the adsorption site first preempting the lower adsorption energy. With the formation of "agglomerated water", there is "dissolved"  $Na<sup>+</sup> [38]$  $Na<sup>+</sup> [38]$  but on the other hand, a wall is formed to prevent diffusion of  $CO<sub>2</sub>$  and shield the force of internal adsorption sites on  $CO<sub>2</sub>$ . Therefore, when  $H<sub>2</sub>O$  and  $CO<sub>2</sub>$ adsorb together, the adsorption capacity of  $CO<sub>2</sub>$  is much lower than that of  $H<sub>2</sub>O$ .

It is believed that the presence of  $H_2O$  accelerates  $CO_2$  adsorption to saturation [\[39,40\]](#page-11-0). However, the mass transfer resistance and shielding effect render the remaining internal O-Si-O sites unable to adsorb  $CO<sub>2</sub>$ . Since H<sub>2</sub>O reacts reversibly with  $CO<sub>2</sub>$  to form unstable H<sub>2</sub>CO<sub>3</sub>, slight adsorption will also occur between  $H_2O$  and  $CO_2$  molecules (binding energy −2.10 kcal/mol) and therefore, a small amount of CO<sub>2</sub> adsorbs onto the outer end of the H2O chain.

#### *3.4. Effects of the NaX skeleton change on CO2 and H2O adsorption*

After observing the adsorption forms of  $CO<sub>2</sub>$  and  $H<sub>2</sub>O$  in NaX, how to regulate the adsorption of  $CO<sub>2</sub>$  and  $H<sub>2</sub>O$  and inhibit the adsorption of H2O is what we want to explore next. Compared with designing zeolites with different pore size types through complex synthesis techniques. We use a simpler temperature effect to change the pore size structure of the zeolite. By discussing the influence of different pore size structures produced by temperature effect on the adsorption of  $CO<sub>2</sub>$  and  $H<sub>2</sub>O$  by NaX zeolite, the key factors affecting the competitive adsorption of  $\mathrm{CO}_2$ and H2O are analyzed.

#### *3.5. Molecular dynamics simulation of changes in the structure of NaX zeolite under temperature effect*

Molecular dynamics (MD) simulation is used to analyze the effects of temperature on the cell parameters such as bond length and bond angle in the NaX zeolite crystal structure. In the range of 50–823.15 K, the unit

<span id="page-5-0"></span>

Fig. 3. Adsorption/desorption mechanism of CO<sub>2</sub> and H<sub>2</sub>O in NaX derived by DSC: (a) CO<sub>2</sub> and (b) H<sub>2</sub>O desorption heat flow and fitted peaks of NaX; (c–e) H<sub>2</sub>O, CO<sub>2</sub>, and CO2-H2O adsorption sites and desorption sequence in NaX.

cell type and lattice constant of NaX do not change in terms of the type and magnitude (Table S3). However, the angle and bond length between O-T-O  $(T = Al, Si)$  change (Table S4). In the range of 433.15–823.15 K, NaX exhibits positive thermal expansion (Fig. 4a,b). NaX shows insignificant negative thermal expansion in the range of 50–300 K as demonstrated previously [\[32\]](#page-10-0).

Further analysis in the 433.15–823.15 K temperature range discloses that the NaX zeolite bond length and bond angle change. Due to the twelve-membered ring framework, the energy in this temperature range is not enough to break the ring framework, and there is no obvious collapse of the framework [\[41\].](#page-11-0) O-Al-O and O-Si-O will restrain each other, resulting in large shrinkage in the porous structure. The average

bond length of  $Al_1$ - $Al_1$  does not change considerably in the temperature range of 50–433.15 K, but fluctuates from 433.15 to 823.15 K (Fig.S6a, b). The average bond length of  $Si<sub>1</sub>-Si<sub>1</sub>$  is stable only between 50 and 100 K, but fluctuates as the temperature increases. Analysis of the angle change of O-T-O  $(T = Al, Si)$  reveals that the bond angle of O-Al-O is relatively stable in the temperature range of 50–773.15 K (Fig.S6c,d). The bond angle of O-Si-O changes greatly in the temperature ranges of 50–200 K and 433.15–673.15 K. The results show that O-Si-O is sensitive to temperature and will produce shrinkage vibrations influenced by temperature, thereby changing the average bond length of  $Si<sub>1</sub>-Si<sub>1</sub>$ .

Since  $Na<sup>+</sup>$  in NaX is not in the skeleton chain, its motion is inconsistent with that of Si and Al. Pure silicalite is not acidic. When



**Fig. 4.** (a)Relative volume variations of NaX at different temperatures simulated by MS; (b) NaX relative density change at different temperatures simulated by MS.

<span id="page-6-0"></span>aluminum atoms are introduced into the zeolite framework to replace silicon atoms, Al is  $3+$  and Si is  $+4$ . Therefore, introduction of each framework aluminum atom generates one negative charge. In order to maintain electrical neutrality, the zeolite framework needs to absorb cations such as  $Na^+$ ,  $K^+$ ,  $Ca^{2+}$ , etc. to balance the charge [\[42\].](#page-11-0)  $Na^+$  is reasonably distributed on the NaX twelve-membered ring with an electrostatic field (positive charge) and Lewis acids and bases (basicity). To maintain charge neutrality in the zeolitic materials, the  $Na<sup>+</sup>$  distribution indicates that the region forms an acidic field or negative charge density  $[43]$ . By monitoring the trajectory of Na<sup>+</sup>, the acidity, alkalinity, and electric field change in the corresponding O-T-O  $(T = AI, Si)$  region can be analyzed. Water preferentially adsorbs at Na<sub>3</sub> sites  $[38]$ . The trajectory of Na<sub>3</sub> fluctuates greatly in the temperature range of 200–673.15 K but less so at other temperatures (Fig. 5a and b). In this temperature range, the acid-base and charge density fields inside the NaX zeolite show obvious changes. Since  $Al^{3+}$  is more basic than  $Si^{4+}$ , O-Al-O (AlO $_4^+$ ) is less acidic than O-Si-O (SiO $_3^{2+}$ ). Therefore, the movement of Na3 is more obvious with changing O-Si-O bond angle (Fig.S6c and d; Table S5). The steric hindrance and interference of the O-T-O  $(T = AI, Si)$ charge field shift the Na<sub>3</sub> distribution angle up and down the O-T-O (T = Al, Si) plane (Fig. 5c and d).

Overall, the temperature effect changes the framework of NaX. Since the temperature change of 0–823.15 K is not enough to destroy the skeleton O-T-O (T = Al, Si) of NaX, O-T-O (T = Al, Si) will produce vibration contraction and stretching consequently affecting the acidity and alkalinity in the vicinity and causing migration of  $Na<sup>+</sup>$ .

Most materials expand as temperature increases, but there are special materials that contract as temperature increases. Materials that shrink with increasing temperature exhibit negative thermal expansion. Zeolites are typical materials with a negative coefficient of thermal expansion [\[44\].](#page-11-0) Previous studies have demonstrated that the thermal behavior of zeolites is determined by the complex interactions between the zeolite framework and cations and water molecules in the channels [45–[47\]](#page-11-0). Commercial zeolites are activated in dry air at low temperature (283.15–423.15 K) during production. This operation not only removes water from the zeolite (wetting water, adsorption water, and a part of crystallization water), but also improves the mechanical strength [\[48\]](#page-11-0). We compared the microstructural changes of NaX zeolite after sintering at 433.15–823.15 K. The temperature change does not affect the macrostructure of NaX zeolite (Figs. S7-9).

#### *3.6. Effect of NaX on CO2 and H2O adsorption performance under temperature effect*

In order to analyze the influence of temperature effect on adsorption performance, the adsorption performance of NaX zeolite sintered at different temperatures for  $CO_2$ ,  $H_2O$  single adsorption and  $CO_2$ - $H_2O$  coadsorption is analyzed. As shown in Fig.  $6(a)$ , when CO<sub>2</sub> and H<sub>2</sub>O are adsorbed alone,  $H<sub>2</sub>O$  exhibits a better adsorption capacity. The original sample has the best adsorption capacity regardless of  $CO<sub>2</sub>$  adsorption



**Fig. 5.** Molecular dynamics simulation of the NaX skeleton changes at different treatment temperatures: (a) and (b) Distance change between Na<sub>3</sub> and Si<sub>1</sub>, A<sub>1</sub>, O<sub>1</sub>; (c) and d) Migration track of  $Na<sub>3</sub>$  on the ring.

<span id="page-7-0"></span>

Fig. 6. Adsorption performance of NaX on CO<sub>2</sub> and H<sub>2</sub>O at different treatment temperatures: (a) Adsorption capacity when CO<sub>2</sub> and H<sub>2</sub>O are adsorbed separately; (b) Adsorption capacity when CO<sub>2</sub> and H<sub>2</sub>O are co-adsorbed; (c) Loss of adsorption capacity when CO<sub>2</sub> and H<sub>2</sub>O are co-adsorbed; (d) Separation ratios of CO<sub>2</sub> and H<sub>2</sub>O during co-adsorption.

 $(6.05 \text{ mmol/g})$  or H<sub>2</sub>O adsorption  $(10.90 \text{ mmol/g})$ . As the treatment temperature increases, the specific surface area on NaX decreases and the overall adsorption capacity decreases (Table S6). The change in the NaX skeleton can modify the adsorption capacity of  $CO<sub>2</sub>$  and H<sub>2</sub>O. For the sample at 773.15 K, the difference between the adsorption capacity of H<sub>2</sub>O and that of CO<sub>2</sub> is the largest, reaching 8.69 mmol/g. [Fig. 5](#page-6-0)(c,d) shows that for the sample at 773.15 K, Na<sub>3</sub> migrate to the inner part of the ring  $(O_2-O_4-Na_3: 137.11°)$ . In the original sample, Na<sub>3</sub> is above the ring (O<sub>2</sub>-O<sub>4</sub>-Na<sub>3</sub>: 54.791°). Considering that adsorption of CO<sub>2</sub> by NaX is mainly Na<sup>+</sup>, as Na<sup>+</sup> migrates into the framework ring, adsorption of  $CO<sub>2</sub>$ is limited due to pore confinement and steric hindrance [\[49\].](#page-11-0) The adsorption energy of NaX to  $H<sub>2</sub>O$  is lower and with the formation of "reunited water", Na<sup>+</sup> can dissolve and migrate and consequently,  $H_2O$ is less restricted.

When low-partial-pressure  $CO<sub>2</sub>$  and  $H<sub>2</sub>O$  are adsorbed by NaX together,  $H_2O$  preempts a large number of  $CO_2$  adsorption sites with faster mass transfer and lower adsorption energy. Therefore, the adsorption capacity of NaX for  $CO<sub>2</sub>$  during co-adsorption is lower than that of single adsorption, and the capacity decay is about 90 % as shown in Fig. 6(b,c). The change in the NaX skeleton has a certain influence on the adsorption capacity of  $CO_2$  when  $CO_2$  and  $H_2O$  are co-adsorbed, but it only increases by 9.4 mmol/g (433.15 K sample). Although the change in the NaX zeolite framework decreases the adsorption capacity of  $CO<sub>2</sub>$ during co-adsorption, it increases and decreases the adsorption capacity of H2O during co-adsorption. The 673.15 K sample also reveals a higher adsorption capacity of NaX for  $H<sub>2</sub>O$  during co-adsorption from 10.25 mmol/g when adsorbed alone to 13.23 mmol/g. This is due to the  $CO<sub>2</sub>$  in NaX, which can adsorb H<sub>2</sub>O and be close to the hydrophobic O-Si-O framework, thereby increasing the distribution space of more  $H_2O$ . As

the treatment temperature increases to 823.15 K, the adsorption capacity of NaX zeolite for co-adsorbed  $H_2O$  decreases by 37.74 % compared to single adsorption. The adsorption capacity of NaX for coadsorbed  $H<sub>2</sub>O$  decreases by 50.5 % in comparison with the pristine sample reaching 5.89 mmol/g. With increasing treatment temperature, NaX zeolite has a greater influence on the adsorption of  $H_2O$  when  $CO<sub>2</sub>$ and H2O are co-adsorbed. This may be due to the sharp increase in the framework angle of O-Al-O, which reduces the force on the surrounding adsorbables  $[50]$ . The movement angle of Na<sup>+</sup> also drops. By modulating the O-T-O (T = Al, Si) structure, the surface energy can be reduced and by controlling the position of  $Na^+$ , the main adsorption sites of  $CO<sub>2</sub>$ and  $H_2O$  and adsorption capacities of  $CO_2$  and  $H_2O$  are affected.

Considering that we want to recover  $CO<sub>2</sub>$  and  $H<sub>2</sub>O$  simultaneously, the separation ratios are derived to evaluate the adsorption selectivity of NaX zeolite for  $CO_2$  and  $H_2O$ , as shown in Fig. 6(d). The sample treated at 433.15 K shows a larger separation ratio from 0.028 (mmol/mmol) to 0.042 (mmol/mmol) indicative of the selectivity of NaX zeolite toward CO2. The sample treated at 523.15 K exhibits a selectivity of NaX zeolite toward H2O of 0.013 (mmol/mmol). Based on the temperature effect on the NaX zeolite framework, Na<sup>+</sup> moves to the farthest point of O-T-O (T  $=$  Al, Si) in the sample treated at 433.15 K.

#### *3.7. Effect mechanism of NaX structural change on CO2 and H2O adsorption*

The effects of temperature on the framework of NaX are assessed by infrared spectroscopy (IR). As shown in Fig.  $7(a)$ , as the treatment temperature increases, the characteristic peaks of the NaX framework gradually weaken [\[51,52\]](#page-11-0), especially symmetric stretching. The peaks

<span id="page-8-0"></span>

**Fig. 7.** Analysis of the skeleton change of NaX influenced by the temperature effects: (a) IR spectra of NaX; (b) XPS Si 2*p* spectra; (c) XPS Al 2*p* spectra; (d) 1 H MAS NMR spectra of NaX after adsorbing  $H_2O$ .

corresponding to asymmetric stretching and the twelve-membered ring framework still exist, indicating that the temperature effect does not break the framework of NaX. However, when the temperature is raised to 823.15 K, the asymmetric stretching peak and twelve-membered ring skeleton weaken. If the temperature goes up further, the framework of NaX breaks and collapses [\[45\]](#page-11-0). This is because the binding energy of O-T-O  $(T = Al, Si)$  decreases due to the temperature effect, as shown in Fig. 7(b,c) and Fig. S10a,b. The change in the bond length and bond angle of O-T-O  $(T = Al, Si)$  affects the electron cloud density in the vicinity consequently altering the acidity and negative charge of O-T-O (T  $=$  Al, Si).

Fig. S4 shows that the change of O-T-O  $(T = Al, Si)$  caused by the temperature effect alters the form of water adsorbed by the NaX zeolite. The samples treated at 433.15 K, 523.15 K, 623.1 5 K, and 723.15 K show small physical adsorption in addition to weak chemical adsorption. This indicates that the temperature-induced changes in the zeolite framework cause weak physical adsorption of water on zeolite O-T-O (T  $=$  Al, Si). Interestingly, the sample treated at 623.15 K has the smallest onset transverse relaxation time (2.0 ms) despite the presence of weakly chemically and physically adsorbed water. The adsorption process gives priority to chemisorption with a higher binding energy, and the desorption process takes priority to physical adsorption with a lower binding energy.

According to the high-frequency NMR <sup>1</sup>H spectra (Fig. 7d), the two  $^{1}$ H peaks shift by different degrees similar to the migration of Na<sup>+</sup>. The <sup>1</sup>H peaks shift by different degrees similar to the migration of Na<sup>+</sup>. The original sample and 823.15 K sample exhibit the similar <sup>1</sup>H peak and

 $Na<sup>+</sup>$  trajectory. However, the <sup>1</sup>H peak of the 573.15 K sample shows a large shift, which is consistent with the shift of the  $Na<sup>+</sup>$  trajectory. Fig. S11-12 show that in the temperature range of 433.15 K-773.15 K, the position of each peak does not change by regulating the O-T-O  $(T =$ Al, Si) skeleton through the temperature effect. However, the peak areas of the  $CO<sub>2</sub>$  and  $H<sub>2</sub>O$  desorption endothermic peaks are affected. This effect is mainly reflected by the desorption heat peak of adsorbed  $H_2O$ . Figs. S11(a) is consistent with [Fig. 3](#page-5-0)(b), showing six main endothermic peaks. The peak around 270 min disappears as the treatment temperature is increased as shown in Fig.  $S11(b-h)$ . When the temperature is higher than 823.15 K, the peak corresponding to agglomerated  $H_2O$ decreases significantly. The adsorption capacity of corresponding coadsorbed H2O also decreases significantly. The pore shrinkage and specific surface area decrease of NaX are the main reasons for the decrease in the free  $H_2O$  adsorption capacity. Fig.  $S12$  shows that the structural change caused by the temperature effect affects the binding energy of CO<sub>2</sub> and H<sub>2</sub>O adsorption. When CO<sub>2</sub> and H<sub>2</sub>O are co-adsorbed, there is only one energy peak for  $CO<sub>2</sub>$  in Fig. S13, while as shown in Fig.  $S14$  (a, c, d, f, g, h), there are multiple  $CO<sub>2</sub>$  adsorption energy peaks, and the peaks start shift to lower energies. These results indicate that the distribution and position of  $Na<sup>+</sup>$  affect the adsorption performance of  $CO<sub>2</sub>$  and H<sub>2</sub>O and by adjusting the zeolite framework, the migration trajectory of  $Na<sup>+</sup>$  can be controlled. The EDS diagrams of NaX zeolite at different temperatures also prove the movement trajectory of Na<sup>+</sup> (Fig. S15).

#### 3.8. CO<sub>2</sub> and H<sub>2</sub>O separation and recovery

Since the captured and separated  $CO<sub>2</sub>$  contains a large amount of desorbed water vapor, it usually needs to be dried and purified before being compressed and stored at a low temperature [\[53\].](#page-11-0) The freezing points of H2O and CO2 are quite different. In order to avoid blockage in the compressor and reduce the compression energy consumption, it is necessary to remove water vapor from  $CO<sub>2</sub>$  to above 99.99 % [\[54\]](#page-11-0). Condensers are used in air separation systems or industrial carbon capture, and external cold sources such as water cooling are used for heat exchange to separate water vapor from  $CO<sub>2</sub>$  [\[55,56\].](#page-11-0) However, increasing the separation rate of water vapor to above 0.9999 requires a high cooling capacity causing energy waste.

The energy consumption of 10 Nm<sup>3</sup>/h of CO<sub>2</sub> streams with different water vapor contents separated by the condenser is calculated. The process flow is shown in Fig. S16, and the energy consumption calculation method refers to formula (1) in SI. After Aspen Plus process simulation, the steady-state operating condenser energy consumption can be obtained on the basis of ensuring component conservation, energy conservation, and compliance with actual operating operations. As shown in Table  $S$ 7, as the purity of the target product  $CO<sub>2</sub>$  goes up from 99 % to 99.99 %, the cooling capacity or cooling water flow rate required to condense water vapor increases. When the water vapor content at the inlet increases, the energy consumption of water vapor separation also increases. For example, drying of 80 % water vapor requires 5 times as much cooling as 1 % water vapor. When NaX coadsorbs  $CO<sub>2</sub>$  and  $H<sub>2</sub>O$ ,  $CO<sub>2</sub>$  adsorbs with less force and desorbs preferentially ([Fig. 3](#page-5-0)d, Fig. S13b). Our previous studies show that there is a difference in the concentration peaks of  $CO<sub>2</sub>$  and  $H<sub>2</sub>O$  in the gas stream after desorption [\[57\]](#page-11-0). For this reason, graded recovery can be realized based on the concentration difference of  $CO<sub>2</sub>$  and  $H<sub>2</sub>O$  in the desorbed gas.

How to further realize the concentration difference graded recovery of CO2 and H2O, we combine the structural change rules of NaX zeolite under the temperature effect and give an outlook on future design based on material design. The design and adjustment of the zeolite structure can control the desorption starting time of  $CO<sub>2</sub>$  and  $H<sub>2</sub>O$ . The pore confinement effect of zeolite can improve the stability of adsorbed molecules by van der Waals and electrostatic interactions, thereby changing the physical and chemical properties of the adsorbed species. In order to ensure the adsorption capacity of zeolite, controlling the range of binding energies can accomplish adsorption stability and low the energy consumption for desorption.

Fig. S14 shows that reducing the surface energy of the NaX zeolite by the temperature effect can reduce the adsorption energy of NaX for  $CO<sub>2</sub>$ 

and  $H_2O$ . Therefore, the energy required for desorption of  $CO_2$  and  $H_2O$ that are weakly chemically/physically adsorbed changes. By controlling the temperature during desorption, the desorption rate of  $CO<sub>2</sub>$  and  $H<sub>2</sub>O$ can be adjusted. Based on the priority desorption sequence of  $CO<sub>2</sub>$  in the desorption process,  $CO<sub>2</sub>$  with a small water vapor concentration is dried and recovered. As shown in Fig. 8(a,b), the desorption time of  $CO<sub>2</sub>$  is approximately 400 s, while the entire desorption process lasts about 1,700 s. This shows that in the  $1/4$  stage of the desorption process,  $CO<sub>2</sub>$ desorption is basically completed. According to the concentrations of the outlet  $CO_2$  and H<sub>2</sub>O, high-purity  $CO_2$  (above 80 %) is dehydrated and recovered. However, low-purity (below 80 %)  $CO<sub>2</sub>$  carries water vapor, and desorption regeneration of the adsorption bed is carried out by the waste heat recovery technique  $[58]$ . When  $CO<sub>2</sub>$  desorption is completed, the remaining desorbed H2O can be directly condensed and recovered with low energy consumption.

#### **4. Conclusions**

We report the adsorption mechanism of  $CO<sub>2</sub>$  and  $H<sub>2</sub>O$  by NaX zeolite, one of the most commercially and industrially applied  $CO<sub>2</sub>$  adsorbents, and discuss its application value in direct air carbon capture. The competitive behavior of  $CO<sub>2</sub>$  and  $H<sub>2</sub>O$  on adsorbent materials helps to design more effective adsorbents, improve the efficiency of carbon capture, and develop more efficient gas separation technologies. The following conclusions can be drawn:

- (a) Due to the limitation of the O-T-O  $(T = Al, Si)$  framework, the temperature effect changes the structure of NaX irregularly. Reducing the surface binding energy of the framework affects the migration of  $Na<sup>+</sup>$  on the twelve rings. The temperature effect reduces the acidity and electron cloud density of the O-T-O ( $T =$ Al, Si) framework, resulting in less  $\text{Na}^+$  being bound by the O-T-O  $(T = Al, Si)$  framework and migration.
- (b) In the temperature range of 433.15–773.15 K, the adsorption capacity of H<sub>2</sub>O during co-adsorption decreases as Na<sup>+</sup> migrates into the ring (the angle of  $O_2-O_4$ -Na<sub>3</sub> increases). The influence of NaX on H2O adsorption during co-adsorption mainly depends on the distribution and position of  $Na^+$ . When the temperature is higher than 773.15 K, the pore size and specific surface of NaX decrease. Steric hindrance and pore confinement are the main reasons for the reduced capacity for  $H_2O$  in the form of agglomerated H<sub>2</sub>O, which is mainly adsorbed by Na<sup>+</sup>.
- (c) During co-adsorption of  $CO<sub>2</sub>$  and  $H<sub>2</sub>O$ , the weak mass transfer and high adsorption energy of  $CO<sub>2</sub>$  limit the distributions of



Fig. 8. Composition changes in the CO<sub>2</sub> and H<sub>2</sub>O co-desorption process: (a) Mass change curve of CO<sub>2</sub> and H<sub>2</sub>O during desorption and (b) Concentration change of  $CO<sub>2</sub>$  in the desorption tail gas.

<span id="page-10-0"></span>adsorption sites. The change in the zeolite structure can reduce the adsorption energy of porous  $CO<sub>2</sub>$ .

We believe that  $Na^+$  sites can be fixed by rational design of the O-T-O  $(T = Al, Si)$  framework [\[59\]](#page-11-0). High-capacity co-adsorption of CO<sub>2</sub> and  $H<sub>2</sub>O$  can be achieved by suppressing the adsorption of  $H<sub>2</sub>O$  or separating the adsorption sites  $[60]$ . CO<sub>2</sub> adsorption on NaX is a physical phenomenon, and the presence of  $H_2O$  fixes a part of  $CO_2$  with weak chemical adsorption. However, owing to the lower adsorption energy and more stable hydrogen bonds,  $H<sub>2</sub>O$  occupies most of the adsorption sites for  $CO<sub>2</sub>$  thus having a shielding effect on the adsorption of  $CO<sub>2</sub>$ . On the contrary,  $CO<sub>2</sub>$  preferentially desorbs during the desorption process. In the co-recovery process of  $CO<sub>2</sub>$  and  $H<sub>2</sub>O$ , separation and recovery of high-purity  $CO<sub>2</sub>$  and  $H<sub>2</sub>O$  can be achieved according to the desorption law of  $CO<sub>2</sub>$  and  $H<sub>2</sub>O$ . Our results disclose that by proper design of the materials and separation technology, joint capture and recovery of CO<sub>2</sub> and H2O by a direct air carbon capture system can be accomplished.

#### **CRediT authorship contribution statement**

**Minghai Shen:** Investigation, Methodology, Writing – original draft, Writing – review & editing. **Fulin Kong:** Data curation, Formal analysis, Writing – review & editing. **Wei Guo:** Methodology, Resources, Software. **Zhongqi Zuo:** Investigation, Resources, Software. **Ting Gao:**  Methodology, Resources, Software. **Sen Chen:** Data curation, Methodology, Resources, Software. **Lige Tong:** Funding acquisition, Supervision, Writing – original draft, Writing – review & editing. **Peikun Zhang:** Methodology, Software, Validation, Visualization. **Li Wang:**  Project administration, Validation, Writing – review & editing. **Paul K. Chu:** Project administration, Visualization, Writing – review & editing. **Yulong Ding:** Visualization, Writing – review & editing.

#### **Declaration of competing interest**

The authors declare that they have no known competing financial interests or personal relationships that could have appeared to influence the work reported in this paper.

#### **Data availability**

No data was used for the research described in the article.

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#### **Appendix A. Supplementary data**

Supplementary data to this article can be found online at [https://doi.](https://doi.org/10.1016/j.cej.2023.147923)  [org/10.1016/j.cej.2023.147923](https://doi.org/10.1016/j.cej.2023.147923).

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# **Impact of H2O on CO<sup>2</sup> Adsorption and Co-Adsorption: Mechanism and High-Performance Adsorbents for Efficient H2O-CO<sup>2</sup> Capture**

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**Table S1**. Characteristics of commercial NaX zeolites.



**Figure S1.** SEM image of the commercial NaX zeolite tested in the experiment.



**Figure S2.** XRD spectra of commercial NaX zeolites used in the experiment.

Except for a few miscellaneous peaks, the pattern is basically consistent with the XRD pattern of the standard Zeolite X (dehydrated) [1]. The few impurity peaks may be due to the addition of a certain amount of adhesive (kaolin) to the commercial zeolite molecular sieves during synthesis [2]. The adhesive is used to produce a certain shape (spherical, bar, etc.) for the zeolite molecular sieve, so that it can be better applied to industrial production. Figure S1 also verifies this phenomenon.



**Figure S3**. Effects of temperature on the porous structure of commercial NaX zeolites and direct air carbon capture based on thermogravimetric and desorption analyses.



**Figure S4.** Transverse relaxation time of H2O adsorption on NaX at different temperatures.



**Table S2**. Adsorption binding energy between molecules.



Figure S5. MS simulation of the respective and common adsorption properties of CO<sub>2</sub> and  $H_2O$  in NaX at low partial pressure: (a) and (b) Adsorption curves of  $CO_2$  and  $H_2O$ ; (c) Diffusion rates of CO<sup>2</sup> and H2O adsorption and co-adsorption; (e) Energy distributions of CO<sub>2</sub> and H<sub>2</sub>O adsorption.

Temperature	a(A)	b(A)	c $(\AA)$	$\alpha$ (°)	$\beta$ (°)	$\gamma$ (°)
50	25.43	25.4	25.41	89.59	90.08	89.77
100	25.42	25.41	25.4	90.01	90.07	90
200	25.55	25.43	25.45	90.31	89.97	89.79
300	25.42	25.53	25.48	90.13	90.51	90.14
433.15(160°C)	25.54	25.53	25.44	90.08	90.13	90.3
$523.15(250^{\circ}\text{C})$	25.5	25.63	25.58	90.04	90.04	89.67
573.15(300°C)	25.6	25.51	25.61	89.99	89.79	89.99
623.15 $(350^{\circ}C)$	25.62	25.62	25.51	89.99	89.87	90.18
673.15(400 $^{\circ}$ C)	25.75	25.79	25.4	90.07	90.34	90.14
723.15(450°C)	25.52	25.72	25.61	90.4	90.66	89.86
773.15(500°C)	25.58	25.66	25.69	90.23	90.2	89.92
823.15(550°C)	25.66	25.65	25.65	90.06	89.82	90.42

**Table S3**. Faujasites lattice parameters of NaX obtained by MD simulations.

Temperature		Bond lengths $(\AA)$		Bond angles( $\circ$ )
(K)	$Si1-Si1$	$Al1-A11$	$O_3 - Al_1 - O_1$	$O1-Si1-O3$
50	11.9276	11.9909	116.879	116.748
100	11.9303	11.9694	115.76	112.855
200	12.0911	11.9856	115.591	118.946
300	12.0242	12.1258	115.632	118.315
433.15(160°C)	11.9697	12.0667	113.97	117.731
523.15(250°C)	12.353	12.1118	112.416	108.993
573.15(300°C)	11.9904	12.3339	114.214	109.732
$623.15(350^{\circ}C)$	11.8491	12.0462	116.155	112.141
673.15(400°C)	12.0849	12.1607	115.024	117.274
723.15(450°C)	12.2578	12.6625	115.22	117.187
773.15(500°C)	12.1015	12.0699	115.478	117.781
823.15(550°C)	12.1122	12.0047	125.33	117.228

**Table S4**. Variation in bond lengths and bond angles of NaX zeolites at different treatment temperatures simulated by MS.



**Figure S6**. Molecular dynamics simulation of the NaX skeleton changes at different treatment temperatures: (a) and (b) Vibrational contraction of  $Si<sub>1</sub>-Si<sub>1</sub>$ , A<sub>l1</sub>-A<sub>l<sub>1</sub>; (c and d)</sub> Angle change of O-T-O (T=Al,Si).



**Table S5.** Trajectories of Na<sup>3</sup> at different treatment temperatures simulated by MS.



**Figure S7**. TG-DSC curves of NaX zeolite at 50-500 ℃.

The main weight loss of NaX zeolite at 50-500 ℃ is caused by adsorbed water and crystal water (crystal frame surface and between crystal frames). As the temperature goes up due to evaporation, the weight loss ratio is 23.8wt%. There is only one heat absorption peak suggesting only water heat absorption.



**Figure S8.** Structure of NaX zeolite treated at different temperatures.

According to the SEM images, commercial 13X is treated at 160 ℃ (pre-dehydration) and 300-550 ℃. The morphology and size of the particles do not change significantly and the structure is stable [3].



**Figure S9.** XRD patterns of NaX zeolite treated at different temperatures.

**Table S6.** Temperature effects on the specific surface area, pore volume, and pore size of NaX zeolite.







**Figure S10.** Skeleton change of NaX at different temperatures: (a) XPS survey spectra

of NaX and (b) XPS Na 1*s* spectra.



Figure S11. Desorption heat flow of CO<sub>2</sub> and H<sub>2</sub>O in NaX at different temperatures.



Figure S12. Fitted peaks of desorption heat flow of CO<sub>2</sub> and H<sub>2</sub>O in NaX at different treatment temperatures.



**Figure S13.** MS simulation of the respective and common adsorption properties of CO<sup>2</sup> and H<sub>2</sub>O in NaX at low partial pressure: (a) Adsorption curves when  $CO<sub>2</sub>$  and H<sub>2</sub>O adsorb together; (b) Energy distributions during co-adsorption of  $CO<sub>2</sub>$  and  $H<sub>2</sub>O$ .



Figure S14. Energy distributions for co-adsorption of CO<sub>2</sub> and H<sub>2</sub>O by NaX at different temperatures simulated by MS.

Figure. S14 shows that the structural changes caused by the temperature effects affect the binding energies of adsorption of CO<sub>2</sub> and H<sub>2</sub>O. During co-adsorption, there is only one energy peak of CO<sup>2</sup> as shown in Figure S10f, while Figures S14a, c, d, f, g, and h reveal a variety of CO<sup>2</sup> adsorption energy peaks.



**Figure S15.** EDS elements of samples treated at different temperatures.

As the contraction vibration of the O-T-O (Al,Si) skeleton is limited as shown in Figure S15, the distribution changes of Al and Si caused by the temperature effect are not obvious. Na acts as a metallic element in the zeolite. Its concentration is relatively small, but the distribution is not uniform. The temperature effect results in migration of  $Na^+$ . However, owing to the limitations of the O-T-O (Al,Si) skeleton and steric hindrance, it does not give rise to a significant density or dispersion. The EDS map of Na in Figure S15 reveals regions where  $Na^+$  is not found (regions with a black color).



Figure S16. Flow diagram of H<sub>2</sub>O and CO<sub>2</sub> condensation recovery system suggested

based on Aspen Plus software.

In AspenTech software such as Aspen Plus or Aspen HYSYS, the calculation of water cooling energy consumption is usually based on thermodynamic principles and the heat balance of the system. The calculation of water cooling energy consumption can be achieved through the following steps:

For water coolers, the energy consumption is usually related to the cooling or heating process of water. Water cooler energy consumption (*Q*) can be calculated by the following formula:

$$
Q = m \cdot C_p \cdot \Delta T \tag{1}
$$

Among them, *m* is the mass flow rate of water (kg/s),  $C_p$  is the specific heat capacity of water (J/(kg·<sup>o</sup>C)), and  $\Delta T$  is the temperature change of water (°C).

This formula describes the energy consumption of cooling or heating water in a water cooler, which absorbs or releases heat by changing the temperature of the water.

This calculation uses industrial circulating cooling water with an inlet temperature of 20°C. The specific heat capacity of water is  $4.2 \times 10^3$  J/(kg·°C).

The outlet temperature of the circulating water changes with the percentage of H2O recovery set in the condensation unit. The condenser module is automatically calculated in Aspen Plus software.

Inlet water	Outlet water			Recirculating
vapor content	vapor content	CO <sub>2</sub> purity	Cold energy	cooling water
$(\%)$	$(\%)$	$(\%)$	(kW)	flow $(kg/s)$
$\mathbf{1}$	$0.01\,$	99.99	$-11.92$	0.5697
$\mathbf{1}$	0.10	99.90	$-10.87$	0.5196
5	0.01	99.99	$-14.23$	0.6802
5	0.10	99.90	$-13.13$	0.6278
5	$\mathbf{1}$	99	$-11.32$	0.5411
10	$0.01\,$	99.99	$-17.11$	0.8183
10	0.10	99.90	$-15.96$	0.7631
10	$\mathbf 1$	99	$-14.1$	0.6742
20	0.01	99.99	$-22.89$	1.095
20	0.10	99.90	$-21.62$	1.034
20	$\mathbf 1$	99	$-19.67$	0.9404
40	$0.01\,$	99.99	$-34.44$	1.647
40	$0.10\,$	99.90	$-32.93$	1.575
40	$\,1$	99	$-30.8$	1.473
60	$0.01\,$	99.99	$-46$	2.199
60	$0.10\,$	99.90	$-44.25$	2.116

**Table S7.** Energy consumption of condensation dehydration for different compositions of  $CO<sub>2</sub>$  and  $H<sub>2</sub>O$ .



## **References**

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